

# Phase Engineering Modulates the Electronic Structure of the IrO<sub>2</sub>/MoS<sub>2</sub> Heterojunction for Efficient and Stable Water Splitting

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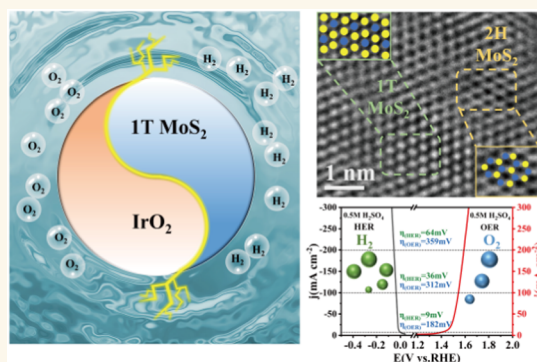


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**ABSTRACT:** The engineering of dual-functional catalytic systems capable of driving complete water dissociation in acidic environments represents a critical requirement for advancing proton exchange membrane electrolyzer technology, yet significant challenges remain. In this work, we investigate an IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure catalyst demonstrating enhanced bifunctional performance for both the oxygen evolution reaction (OER) and hydrogen evolution reaction (HER) under acidic conditions. Strategic incorporation of IrO<sub>2</sub> into the MoS<sub>2</sub>/CNT heterojunction induces a partial phase transformation from 2H to the metastable 1T configuration in MoS<sub>2</sub>, thereby modulating the electronic structure of IrO<sub>2</sub> and improving the catalytic performance for overall water splitting. The optimized IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst exhibited exceptional overpotentials of 9 mV (HER) and 182 mV (OER) at a current density of 10 mA cm<sup>-2</sup> in acidic media. Full-cell evaluations further confirmed its practical potential, showing a 1.47 V operation voltage that outperforms standard Pt/C||IrO<sub>2</sub> counterparts by 120 mV. The experimental results revealed that the n–n heterojunction between IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT generates a built-in electric field, enhancing charge redistribution and electron transport. Moreover, density functional theory simulations further identify iridium centers as dominant catalytic loci, with a metastable 1T-MoS<sub>2</sub> phase mediating charge equilibration at atomic interfaces. This modification facilitates \*OH adsorption and \*OOH deprotonation and lowers the kinetic barrier during the water-splitting process.



## 1. INTRODUCTION

Advancing efficient and scalable water-splitting systems represents a critical pathway for tackling pressing global energy demands and environmental sustainability issues. Proton exchange membrane (PEM) electrolysis has emerged as a pivotal technology for green hydrogen generation due to its superior energy conversion rates, modular architecture, and dynamic operational responsiveness.<sup>1,2</sup> Nevertheless, kinetic limitations in both the anodic oxygen evolution reaction (OER) and cathodic hydrogen evolution reaction (HER) processes continue to hinder progress, necessitating reliance on rare noble metal-based catalysts like IrO<sub>2</sub> and Pt that suffer from supply constraints and high costs.<sup>3</sup> These challenges underscore the urgent need for innovative bifunctional electrocatalyst designs that demonstrate optimized reaction efficiency, economic viability, and acid-stable performance to facilitate modular PEM electrolyzer system development.

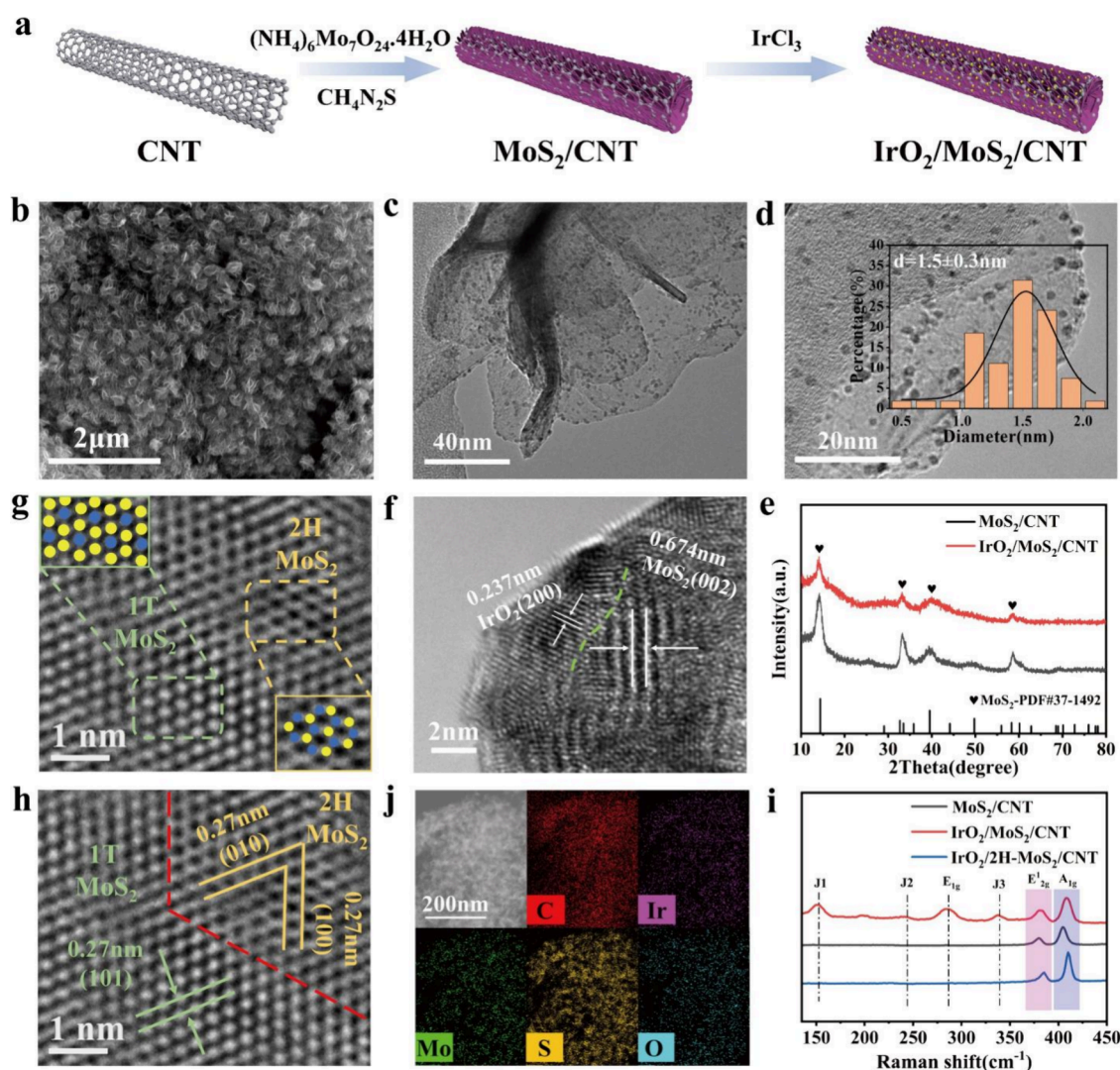
Implementing heterojunction structural engineering provides a sophisticated methodology to enhance bifunctional electrocatalytic capabilities and significantly reduce reliance on

high-cost metallic components. Recently, MoS<sub>2</sub>, a widely studied transition metal dichalcogenide, has shown great potential as an HER catalyst due to its earth abundance, tunable electronic structure, and high catalytic activity. In its natural state, MoS<sub>2</sub> typically exists in the semiconducting 2H phase, which exhibits limited HER activity due to poor conductivity and insufficient active sites.<sup>4</sup> Interestingly, the metallic 1T phase of MoS<sub>2</sub> demonstrates significantly enhanced conductivity and catalytic activity, improving its catalytic activity.<sup>5–9</sup> Unfortunately, the metastable 1T phase of MoS<sub>2</sub> presents significant synthesis challenges, being neither

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**Figure 1.** Synthesis and structural characterizations of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT. (a) Schematic illustration for the preparation of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, (b) SEM, (c) TEM, (d) HRTEM, (e) XRD, (f, g and h) TEM, (i) Raman spectra of MoS<sub>2</sub>/CNT, IrO<sub>2</sub>-MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT. (j) STEM images of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT.

naturally obtainable nor readily producible at scale through conventional chemical or physical synthesis routes.<sup>10,11</sup>

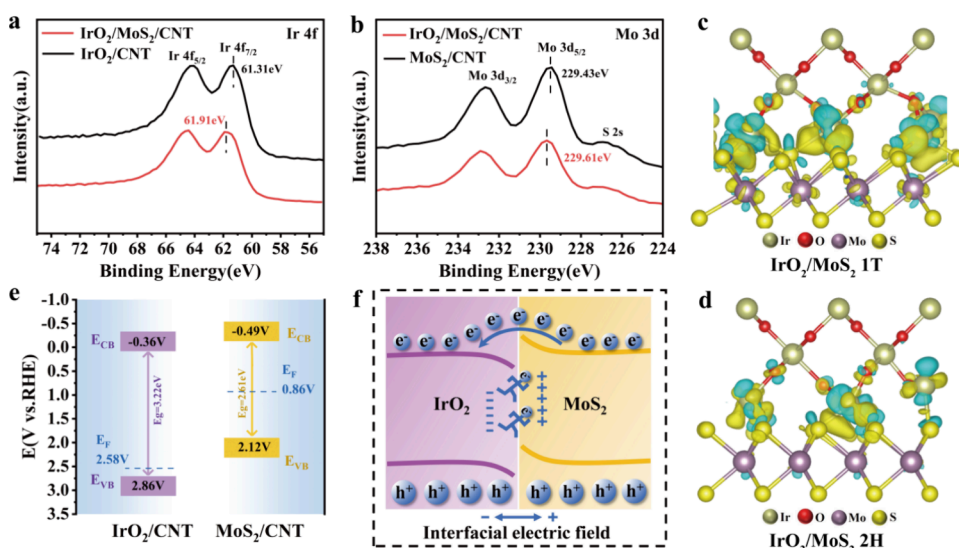
Herein, we report the construction of an IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterojunction, where the phase of MoS<sub>2</sub> is partially transformed from the 2H phase to the 1T phase. The introduction of IrO<sub>2</sub> not only provides highly active sites for the OER but also facilitates the phase transition of MoS<sub>2</sub> through strong interfacial interactions. Additionally, carbon nanotubes (CNTs) are incorporated to enhance the overall conductivity and structural stability of the heterojunction. The synergistic effect between IrO<sub>2</sub> and 1T MoS<sub>2</sub>, combined with the conductive CNT network, enables efficient charge redistribution and electron transfer, thereby significantly improving the bifunctional activity for both HER and OER. The optimized IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst achieved remarkable performance with low overpotentials of 9 mV (HER) and 182 mV (OER) at 10 mA cm<sup>-2</sup> under an acidic environment, while requiring only 1.47 V to drive 10 mA cm<sup>-2</sup> in a two-electrode configuration, outperforming commercial Pt/C||IrO<sub>2</sub> counterparts by 120 mV in cell voltage. Combining the experimental and DFT results demonstrated that Ir sites function as primary active centers, with 1T-MoS<sub>2</sub> enhancing charge redistribution and interfacial

structure to promote \*OH adsorption, \*OOH deprotonation, and lower kinetic barriers during water splitting.

## 2. RESULTS AND DISCUSSION

**2.1. Preparation and Characterization of IrO<sub>2</sub>/MoS<sub>2</sub>/CNTs.** The synthesis procedure for the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure is schematically depicted in Figure 1a. Briefly, the heterostructure IrO<sub>2</sub>/MoS<sub>2</sub>/CNT is prepared via a two-step hydrothermal method. Initially, MoS<sub>2</sub>/CNT composites were synthesized by reacting molybdic acid with thiourea in the presence of carbon nanotubes at 180 °C for 24 h. Subsequent incorporation of IrCl<sub>3</sub>·xH<sub>2</sub>O into the above precursor followed by thermal treatment at 100 °C for 10 h, marked as IrO<sub>2</sub>/MoS<sub>2</sub>/CNT. Morphological characterization via scanning electron microscopy (SEM) revealed that the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT (Figure 1b) composite maintains the nanosheet architecture inherited from its MoS<sub>2</sub>/CNT precursor (Figure S1a). Transmission electron microscopy (TEM) analysis (Figure 1c) further demonstrated the homogeneous dispersion of IrO<sub>2</sub> nanoparticles within the matrix, exhibiting an average particle size of 1.5 nm (Figure 1d), compared with the MoS<sub>2</sub>/CNT precursor (Figure S1b)





**Figure 2.** (a) XPS spectra of Ir 4f of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/CNT; (b) XPS spectra of Mo 3d of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT; the distribution of the IrO<sub>2</sub>/MoS<sub>2</sub> 1T (c) and IrO<sub>2</sub>/MoS<sub>2</sub> 2H (d); (e) energy band structure alignments of IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT; and (f) schematic of the interface electronic structure in the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterojunction.

and IrO<sub>2</sub>/CNT with uneven particles (Figure S2). The crystalline phase of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures was characterized using X-ray diffraction (XRD). The XRD pattern confirmed the preservation of MoS<sub>2</sub>'s hexagonal crystal structure (PDF#37–1492) in the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT composite (Figure 1e).<sup>5</sup> However, the absence of distinct IrO<sub>2</sub> diffraction peaks suggested effective nanoparticle dispersion and subcritical crystallite dimensions below XRD detection limits, as evidenced by comparison with IrO<sub>2</sub>/CNT reference samples (Figure S3). High-resolution TEM (HRTEM) imaging further confirmed the measured interplanar spacings of 0.237 and 0.674 nm correspond to the (200) facets of rutile IrO<sub>2</sub> and (002) facets of MoS<sub>2</sub> in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterojunction (Figure 1f).<sup>5,12</sup> Interestingly, two distinct lattice structures were observed: hexagonal lattice regions with ~0.27 nm spacing (corresponding to (100)/(010) planes of 2H-phase MoS<sub>2</sub>) and trigonal lattice regions with 0.27 nm fringe spacing (attributed to (101) facets of 1T-phase MoS<sub>2</sub>),<sup>5,13</sup> as given in typical HRTEM images (Figure 1g,h), indicating that the original 2H MoS<sub>2</sub> is partially converted to 1T MoS<sub>2</sub> following IrO<sub>2</sub> introduction.

The coexisting 1T and 2H MoS<sub>2</sub> phases in the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures were further confirmed by Raman spectroscopy. The Raman spectrum showed three distinct signals at 150 cm<sup>-1</sup> (J<sub>1</sub>), 218 cm<sup>-1</sup> (J<sub>2</sub>), and 327 cm<sup>-1</sup> (J<sub>3</sub>), corresponding to the S–Mo–S bond vibrations in the 1T-MoS<sub>2</sub> phase (Figure 1i), according to previous reports on 1T-MoS<sub>2</sub>.<sup>5,14</sup> These peaks were absent in the Raman spectra of the 2H MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT (synthesized by the physical mixing method), clearly indicating the successful stabilization of the metastable 1T phase of MoS<sub>2</sub> within the heterostructure. The elemental distribution within the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure was further characterized by scanning transmission electron microscopy (STEM) elemental mapping (Figure 1j). The elemental maps demonstrated a uniform spatial distribution of C, Mo, Ir, O, and S components across the heterostructure, confirming structural homogeneity and compositional integration. These collective findings confirm the successful formation of coexisting 1T and 2H-

MoS<sub>2</sub> phases in the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures achieved through IrO<sub>2</sub> incorporation.

The chemical state and atomic structure of the heterostructured IrO<sub>2</sub>/MoS<sub>2</sub>/CNT composite were explored by XPS.<sup>12,15,16</sup> The XPS survey for the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT (Figure S4) indicates the presence of Ir, Mo, O, C and S. Compared to IrO<sub>2</sub>/CNT, IrO<sub>2</sub>/MoS<sub>2</sub>/CNT of the Ir 4f peak is positively shifted by 0.6 eV (Figure 2a); on the other hand, the Mo 3d<sub>5/2</sub> and Mo 3d<sub>3/2</sub> signal in the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures exhibited a 0.18 eV positive shift relative to the bare MoS<sub>2</sub>/CNT (Figure 2b), indicating strong electronic interactions between IrO<sub>2</sub> and MoS<sub>2</sub>, which confirm heterostructure interface formation.

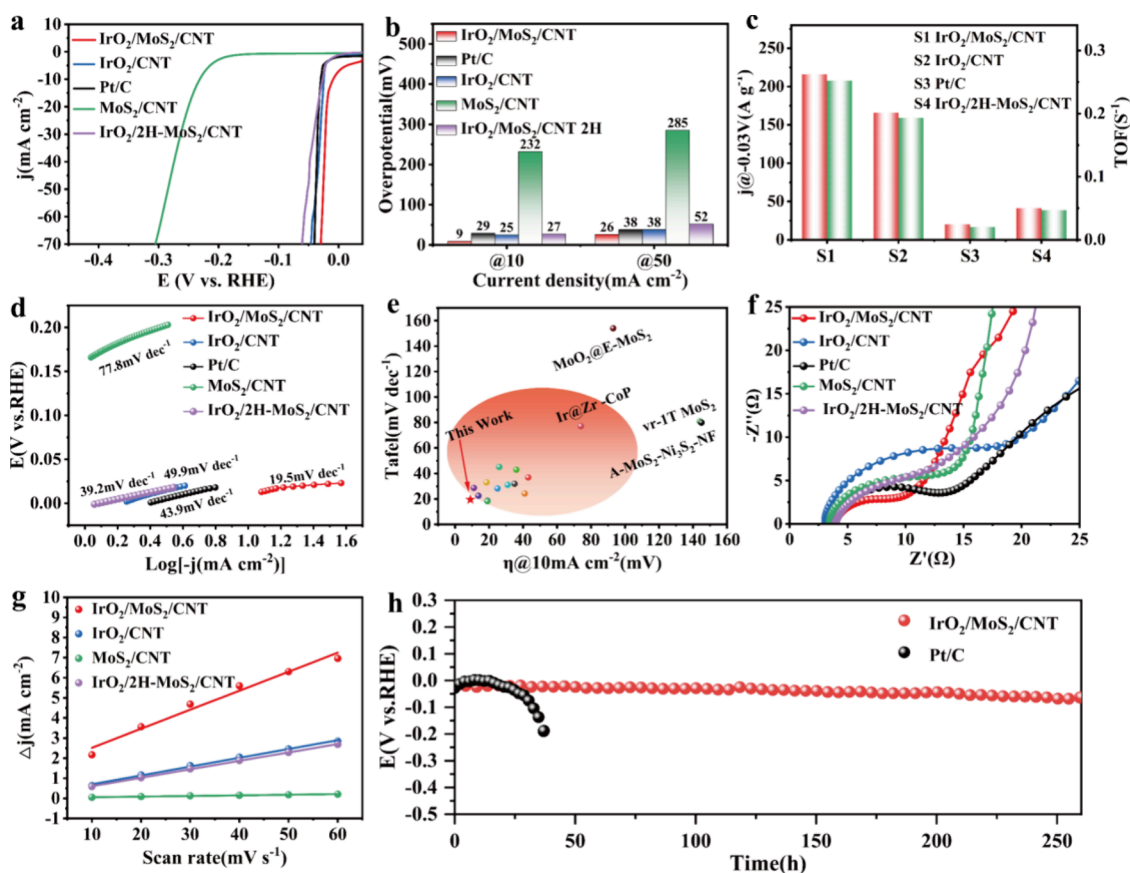
Differential charge density analysis further revealed electron accumulation (yellow regions) and depletion (blue regions) patterns. As illustrated in Figure 2c, notable charge accumulation at the MoS<sub>2</sub> interface suggests electron transfer from IrO<sub>2</sub> to MoS<sub>2</sub>, compared to the IrO<sub>2</sub>/MoS<sub>2</sub> 2H model (Figure 2d). The Bader charge values for the Ir atom in the IrO<sub>2</sub>/MoS<sub>2</sub> 1T structure were determined to be 8.32 e (Table S1), confirming significant electron transfer from IrO<sub>2</sub> to MoS<sub>2</sub> and highlighting charge redistribution within the IrO<sub>2</sub>/MoS<sub>2</sub> system.<sup>17–20</sup> The constructed IrO<sub>2</sub>/MoS<sub>2</sub> catalyst achieved charge density redistribution through phase interface engineering with electron-deficient Ir surface sites serving as active centers. This configuration effectively modulates intermediate adsorption behavior and reduces reaction barriers.

The electron transfer at the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterojunction interface was comprehensively investigated through UV–vis spectroscopy (Figure S5a), Mott–Schottky plots, and valence band characterization.<sup>21–23</sup> The optical band gap energy (E<sub>g</sub>) was calculated using the fundamental equation:

$$(\alpha h\nu)^n = k(h\nu - E_g)$$

in which  $h\nu$  represents photoenergy,  $\alpha$  denotes the absorption coefficient, and  $k$  is a material-dependent constant. The band gap energies of IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT were calculated as 3.22 and 2.61 eV, respectively (Figure S5b).

The Mott–Schottky plots in Figure S6 reveal that both IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT exhibit characteristic n-type



**Figure 3.** (a) LSV curves of as-grown catalysts in 0.5 M H<sub>2</sub>SO<sub>4</sub>; (b) Relationship between overpotential and the catalysts with different current density; (c) Mass activity of different catalysts at −0.03 V and TOF at the overpotential of 30 mV; (d) Tafel plots calculated from Figure 3a; (e) Comparison of Tafel slope and overpotentials at the current density of 10 mA cm<sup>−2</sup> between IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and recently reported Mo-based electrocatalysts; (f) Nyquist plots for the different electrodes; (g) Charging current density differences plotted versus scan rate; (h) Long-term electrochemical stability of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and Pt/C was measured at the current density of 10 mA cm<sup>−2</sup>.

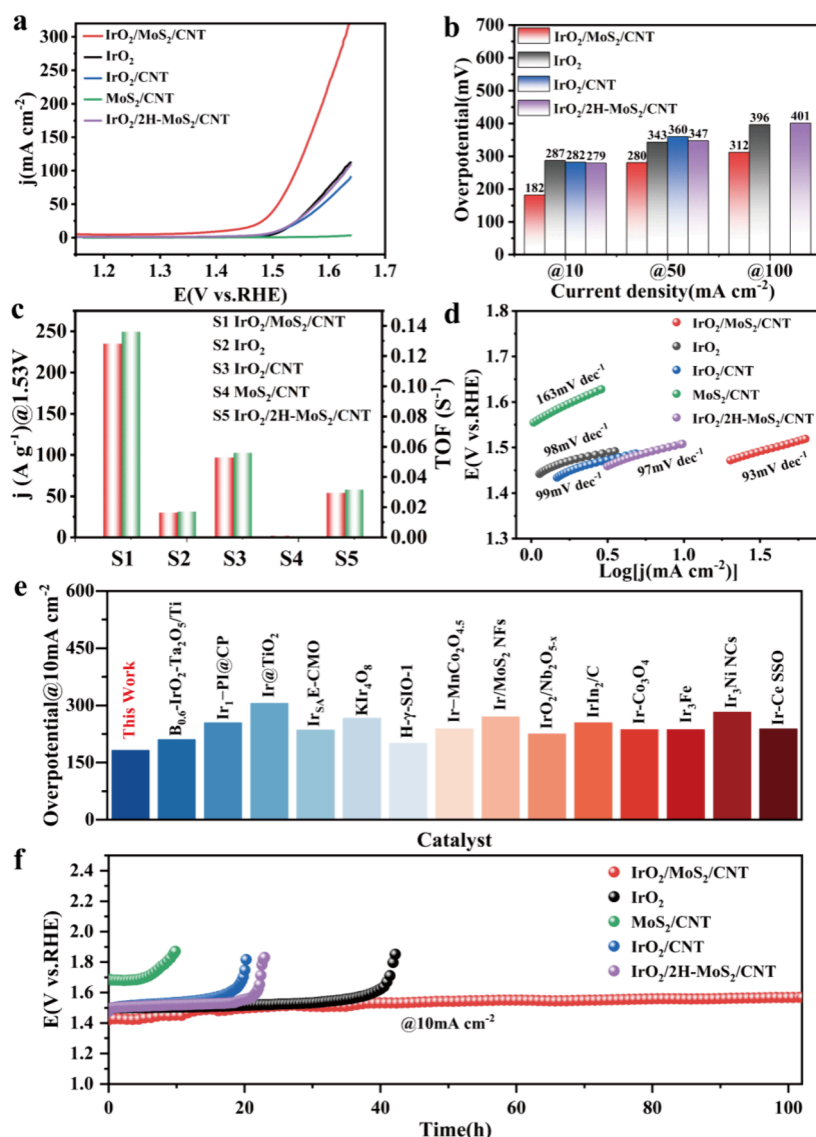
semiconductor behavior, evidenced by their positive linear slopes.<sup>21,22</sup> The flat band ( $E_{fb}$ ) potentials were measured at −0.59 V (IrO<sub>2</sub>/CNT) and −0.72 V (MoS<sub>2</sub>/CNT) vs Ag/AgCl. The conduction band (CB) minimum positions were established at −0.36 vs reversible hydrogen electrode (RHE) for IrO<sub>2</sub>/CNT and −0.49 V vs RHE for MoS<sub>2</sub>/CNT, consistent with n-type semiconductor behavior. The valence band (VB) maxima were determined to be 2.86 and 2.12 V vs RHE for IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT, respectively, through the combined band gap and electrochemical analysis. Energy band diagrams (Figure 2e) were constructed using the established  $E_g = E_{CB} - E_{VB}$ , illustrating the relative band positions of IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT. Valence band XPS measurements (Figure S7) revealed Fermi level positions at 2.58 eV for IrO<sub>2</sub>/CNT and 0.86 eV for MoS<sub>2</sub>/CNT relative to their valence band maxima. Heterojunction formation enables spontaneous electron migration from the higher-Fermi-level MoS<sub>2</sub>/CNT component to the lower-Fermi-level IrO<sub>2</sub>/CNT. This interfacial charge transfer triggers band bending and spatial charge redistribution, as depicted in Figure 2f.

The above results, combined with the HRTEM image of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT in Figure 1f, confirm the successful formation of an n–n heterojunction structure in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT. The established heterojunction between IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT generates a built-in electric field that enhances charge carrier mobility across the interface. The surface potential modification creates positive charge centers on IrO<sub>2</sub>/CNT,

promoting OH<sup>−</sup> species electron donation. The transferred electrons subsequently undergo recombination with valence band hole carriers within the IrO<sub>2</sub>/CNT lattice, facilitated by the heterojunction-induced charge compensation mechanism. The optimized interfacial electronic structure significantly improves the charge transfer kinetics and modifies the surface electronic configuration, thereby enhancing the electrocatalytic activity of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT material for water splitting.

**2.2. Catalytic Activity for HER and OER.** The electrocatalytic performance of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst for both HER and OER was systematically investigated in 0.5 M H<sub>2</sub>SO<sub>4</sub> with a three-electrode configuration. The HER activity was evaluated through linear sweep voltammetry (LSV) measurements. The IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst exhibited enhanced HER activity compared to bare MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/CNT (Figures 3a and S8). The optimal IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst showed small overpotentials of 9 and 26 mV at current densities of 10 and 50 mA cm<sup>−2</sup>, respectively, outperforming commercial Pt/C catalyst (29 and 38 mV), as given in Figures 3b and S9. As the current density increased, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst maintained a greater advantage in terms of low overpotential compared with the other samples. In comparison, the catalysts synthesized by the physical mixing method, namely, the IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT, exhibited lower catalytic performance compared to the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT with the 1T-2H phase, suggesting enhanced electronic structure modulation through phase engineering, thereby enhancing the



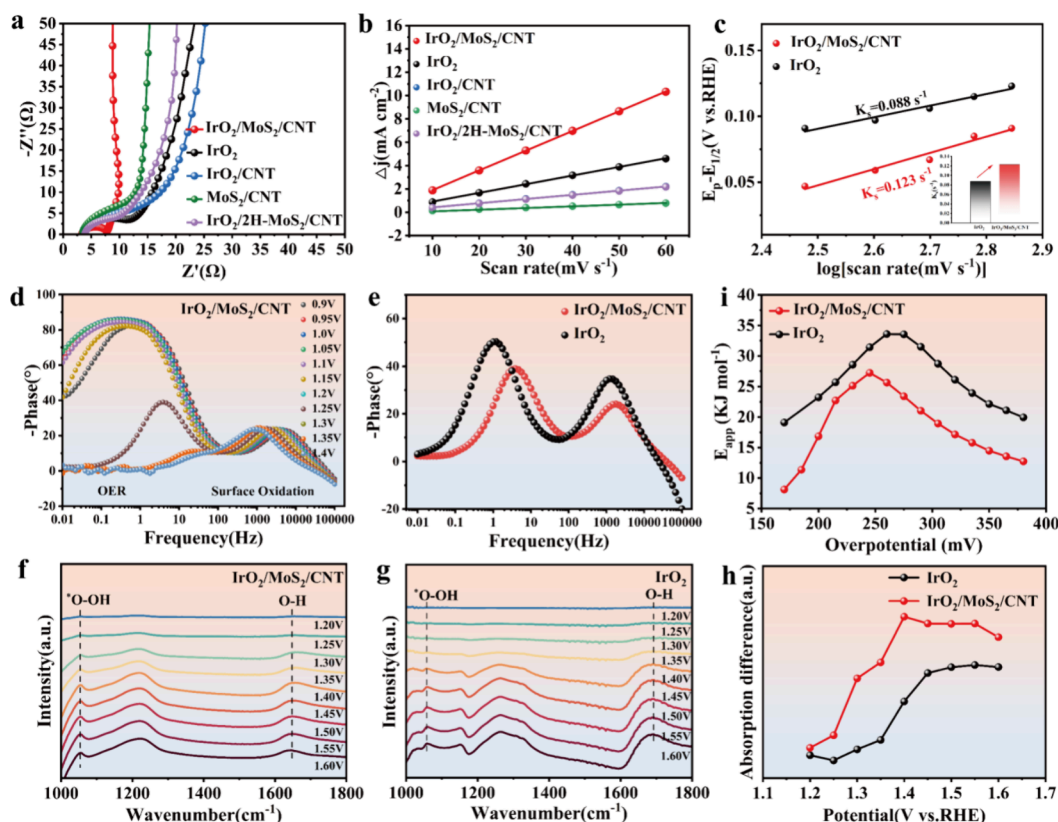


**Figure 4.** (a) LSV curves for different catalysts with 90%-iR correction in acid medium; (b) relationship between overpotential and the catalysts with different current density; (c) mass activity of different catalysts at 1.53 V and TOF at the overpotential of 300 mV; (d) Tafel plots calculated from Figure 2a; (e) comparison of overpotentials at the current density of 10 mA cm<sup>-2</sup> between IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and recently reported Ir-based electrocatalysts; (f) long-term electrochemical stability test was measured at the current density of 10 mA cm<sup>-2</sup>.

activity and kinetic performance of the catalyst. Mass activity measurements at  $-0.03$  V vs RHE showed IrO<sub>2</sub>/MoS<sub>2</sub>/CNT achieving 216 A g<sup>-1</sup>, representing 10.8-fold and 5.2-fold improvements over Pt/C (20 A g<sup>-1</sup>) and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT (41.2 A g<sup>-1</sup>) electrodes, respectively (Figure 3c). The turnover frequency (TOF) of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst was 0.252 s<sup>-1</sup> at 30 mV overpotential, exceeding Pt/C (0.02 s<sup>-1</sup>) and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT (0.047 s<sup>-1</sup>) by factors of 12.6 and 5.4, respectively, evidencing optimized catalytic site exposure efficiency and enhanced fundamental catalytic performance. Tafel slope analysis revealed accelerated HER kinetics with IrO<sub>2</sub>/MoS<sub>2</sub>/CNT (19.5 mV dec<sup>-1</sup>), significantly lower than that with MoS<sub>2</sub>/CNT (77.8 mV dec<sup>-1</sup>), IrO<sub>2</sub>/CNT (49.9 mV dec<sup>-1</sup>), and Pt/C (43.9 mV dec<sup>-1</sup>), as depicted in Figure 3d. In fact, the catalytic activities of the as-fabricated IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst are comparable with recently reported analogous electrocatalysts in acid electrolytes,<sup>24–26</sup> as presented in Figure 3e and Table S2.

These HER performance enhancements were quantitatively validated through complementary electrochemical impedance spectroscopy (EIS) and electrochemically active surface area (ECSA) analyses. Evidently, the Nyquist plot of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures exhibits the smallest semicircle diameter among all of the tested electrodes, suggesting minimal interfacial charge transfer resistance (Figure 3f). Meanwhile, the contact angle between water and IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructures showed an anticipated decrease compared with 2H MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/CNT, because of the good hydrophilicity of 1T MoS<sub>2</sub> (Figure S10). This enhanced wettability promotes optimal electrolyte-catalyst interfacial contact while maximizing the active site accessibility. To further analyze the intrinsic catalytic activity, CV-based ECSA evaluation (Figure S11) revealed superior active site exposure in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, as evidenced by its steeper linear fitting slope compared to control samples (Figure 3g).

Besides its high activity and fast kinetics, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure also demonstrated remarkable opera-



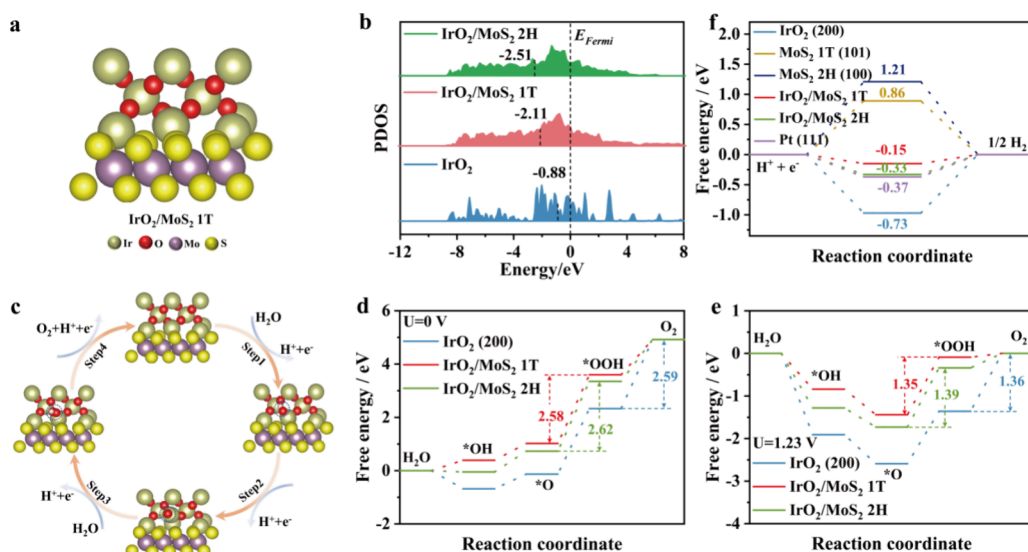
**Figure 5.** (a) EIS curves; (b) Charging current density differences plotted versus scan rate. (c) Ks of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub>; (d) Bode phase plots for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalysts at different applied potentials versus SCE. in 0.5 M H<sub>2</sub>SO<sub>4</sub>; (e) Bode phase plots of pure IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub> at 1.3 V vs SCE. In situ FTIR spectra of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT (f) and IrO<sub>2</sub> (g); (h) Peak intensity enhancements of the absorption bands at 1055 cm<sup>-1</sup> as a function of applied potential; (i) Arrhenius plots of the kinetic current illustrating different activation energy of OER at 1.5 V versus RHE for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub>.

tional stability in acid media, maintaining negligible potential deviation during prolonged operation at 10 mA cm<sup>-2</sup> for 200 h (Figure 3h). Conversely, the Pt/C catalyst showed a sharp potential decrease within the initial 40 h range. Additionally, stepwise chronopotentiometric testing of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode was conducted across current densities spanning 10–50 mA cm<sup>-2</sup> (Figure S12). Remarkably, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode exhibited complete potential recovery to baseline values at 10 mA cm<sup>-2</sup> following high-current-density operation. This behavior highlights the mechanical robustness and stability of the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode under dynamic operating conditions.

The OER evaluation in 0.5 M H<sub>2</sub>SO<sub>4</sub> revealed enhancement trends analogous to those of HER performance (Figure S13). Among the LSV curves shown in Figure 4a, for the OER process, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrocatalysts exhibited significantly higher electrocatalytic activities compared with other electrocatalysts. The IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure exhibited superior electrocatalytic performance with an overpotential of 182 mV at 10 mA cm<sup>-2</sup>, demonstrating significant advantages over both IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT (279 mV) and commercial IrO<sub>2</sub> catalysts (287 mV). This performance gap became even more pronounced at elevated current densities, where the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterostructure maintained a low overpotential of 312 mV at 100 mA cm<sup>-2</sup>, outperforming its 2H-phase counterpart (401 mV) and commercial IrO<sub>2</sub> (396 mV) by substantial margins, as illustrated in Figure 4b.

Mass activity analysis revealed 235 A g<sup>-1</sup> at 1.53 V vs RHE for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst, representing approximately 7.9-fold and 4.4-fold improvements over commercial IrO<sub>2</sub> (29.8 A g<sup>-1</sup>) and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT electrodes (54 A g<sup>-1</sup>), respectively (Figure 4c). TOF calculations at 300 mV overpotential showed 0.136 s<sup>-1</sup> for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, exceeding 8-fold and 4.3-fold improvements over commercial IrO<sub>2</sub> (0.017 s<sup>-1</sup>) and IrO<sub>2</sub>/2H-MoS<sub>2</sub>/CNT (0.0315 s<sup>-1</sup>) respectively, confirming superior atomic-level efficiency in oxygen evolution catalysis. The catalytic enhancement of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT was quantitatively confirmed through Tafel analysis, recording a 93 mV dec<sup>-1</sup> slope that demonstrates accelerated oxygen evolution kinetics compared to those of commercial IrO<sub>2</sub> electrodes (Figure 4d). These results suggest that the 1T-2H phase of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT facilitates efficient water adsorption and improves reaction kinetics due to its electronic structure rearrangement and enhanced hydrophilicity. At 10 mA cm<sup>-2</sup>, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT exhibited overpotentials competitive with or superior to reported high-performance Ir-based catalysts, as validated by comparative literature analysis (Figure 4e and Table S3).<sup>27–30</sup> Furthermore, the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT catalyst demonstrated exceptional stability, maintaining performance at 10 mA cm<sup>-2</sup> for 100 h, outperforming other catalysts (Figure 4f).

**2.3. Revealing the Origin of OER Activity.** The enhanced OER activity of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT was systematically investigated by using comprehensive electrochemical characterizations. First, Nyquist plots from EIS at open-circuit potential revealed the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode exhibiting



**Figure 6.** (a) Structure model of IrO<sub>2</sub>/MoS<sub>2</sub> 1 T; (b) PDOS of different catalysts; (c) key intermediate adsorption configuration models for the Ir site of IrO<sub>2</sub>/MoS<sub>2</sub> 1 T; (d and e) computed overpotential versus different catalysts for OER at  $U = 0$  and  $U = 1.23$  V; (f) Computed overpotential versus different catalysts for HER.

the smallest charge-transfer impedance and largest Warburg impedance among tested catalysts (Figure 5a).<sup>31</sup> This indicates optimized electron transfer kinetics and enhanced mass transport properties, which can be attributed to the synergistic 2H-1T phase transition-induced electronic modulation in MoS<sub>2</sub>. Next, the ECSA was evaluated via double-layer capacitance ( $C_{dl}$ ) according to CVs (Figure S14). The slope of the linear fit for the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode was steeper compared to the other electrodes (Figure 5b), indicating that the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode possesses superior active site exposure.

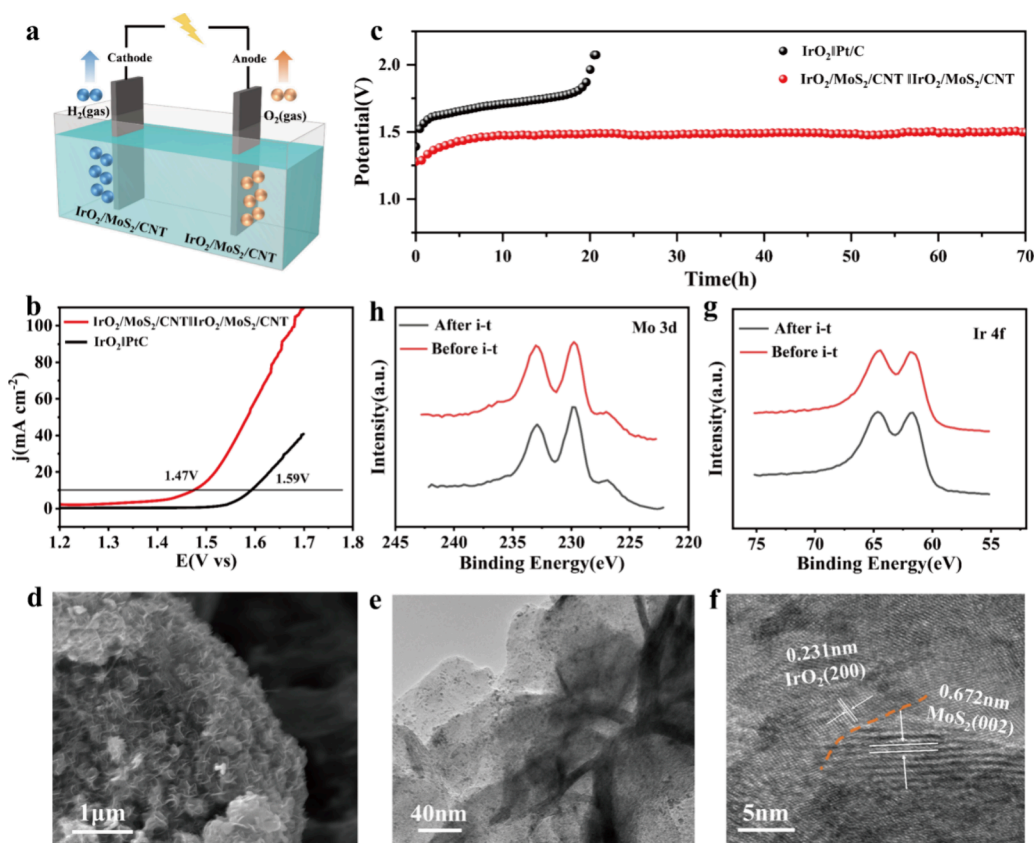
The OER reactivity is governed by optimal adsorption energetics of oxygenated intermediates (OH\*, O\*, OOH\*) at catalytic metal centers.<sup>32</sup> Laviron analysis quantified \*OH binding affinity at Ir active sites in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT versus commercial IrO<sub>2</sub>.<sup>16</sup> Linear correlation between anodic peak currents and the square root of the CVs scan rate confirmed diffusion-controlled \*OH adsorption kinetics (Figure S15). The calculated rate constant ( $K_s$ ) for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT was 0.123 s<sup>-1</sup>, which is higher than that of commercial IrO<sub>2</sub> (0.088 s<sup>-1</sup>), as quantified in Figure 5c. This enhanced kinetic parameter confirms stronger \*OH intermediate stabilization at the interfacial active sites of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT. Competitive methanol oxidation assays probed OH\*/O\* binding energetics in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, further revealing optimized intermediate-specific adsorption through electrophilicity-dependent competitive adsorption (Figure S16).<sup>33</sup>

Mechanistic investigation of oxygen evolution kinetics employed operando electrochemical impedance spectroscopy to track oxygenated intermediate adsorption dynamics on the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT during catalytic operation. Figures 5d and S17 showcase characteristic interfacial responses through Nyquist and Bode phase plots recorded under operando conditions (0.9–1.4 V vs SCE), benchmarked against pure IrO<sub>2</sub> (Figure S18). At 1.25 V, the Bode phase plots showed that the low-frequency interfacial signature of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT had reduced phase angles and characteristic frequency shift compared to IrO<sub>2</sub> (Figure 5e), indicating faster deprotonation of \*OOH intermediates.<sup>34,35</sup>

To further investigate the catalytic mechanism, operando FTIR spectroscopy employing a specialized electrochemical cell tracked oxygen intermediate evolution during catalysis.<sup>36,37</sup> Figure 5f presents potential-dependent FTIR spectra for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, revealing a band in the range of 1600–1800 cm<sup>-1</sup> corresponding to the H–O–H bending mode ( $\delta$ H–O–H) of water. Progressive potential application induced a characteristic 1055 cm<sup>-1</sup> vibration signature in IrO<sub>2</sub>/MoS<sub>2</sub>/CNT, diagnostic of \*OOH intermediate formation during the process of acidic OER.<sup>36</sup> Control experiments with pure IrO<sub>2</sub> showed no detectable features in the 1000–1100 cm<sup>-1</sup> range (Figure 5g). This spectroscopic verification confirms that IrO<sub>2</sub>/MoS<sub>2</sub>/CNT heterointerface engineering enhances 4e<sup>-</sup> pathway selectivity while stabilizing critical oxygen intermediates. Potential-dependent intensification of the 1055 cm<sup>-1</sup> vibrational mode (Figure 5h) quantitatively tracks \*OOH accumulation dynamics during the OER operation. Notably, oxygen coupling events preceding \*OOH formation were detected at 1.25 V, indicating sequential intermediate transformation kinetics in the catalytic cycle. Furthermore, complementary rotating ring-disk electrode (RRDE) analysis demonstrated minimal peroxide byproduct formation, confirming that the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode predominantly followed 4-electron pathway selectivity through RRDE quantification, as shown in Figure S19.<sup>38</sup> This mechanistic synergy between thermodynamic \*OOH stabilization and kinetic oxygen coupling enables optimized four-electron pathway efficiency and enhanced reaction kinetics in acidic media.<sup>37</sup>

The experimental conclusions received additional substantiation through analysis of the derived activation energy ( $E_a$ ) values.<sup>16,39</sup> Kinetic studies under varying temperature conditions were conducted for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub> electrodes to establish their respective  $E_a$  values during the acidic OER. LSV measurements of both electrode systems in 0.5 M sulfuric acid electrolyte were obtained across temperatures ranging from 20 to 60 °C (Figure S20). Consistent with thermodynamic principles, both catalytic systems demonstrated progressively enhanced activity with rising operational temperatures. Through the application of the Arrhenius





**Figure 7.** (a) Acid electrolyzer using two identical  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  as both the cathode and anode; (b) LSV curves using  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  as two electrodes; (c) chronopotentiometric response curve of  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  at  $10 \text{ mA cm}^{-2}$ ; SEM (d) and TEM images (e, f) and XPS (g, h) of  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  after the stability test.

relationship, activation energy values at 1.5 V versus RHE were determined as shown in Figure 5i. The calculated  $E_a$  for the  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  electrode measured  $23.4 \text{ kJ mol}^{-1}$ , exhibiting a substantial reduction compared to that of pure  $\text{IrO}_2$  ( $33.6 \text{ kJ mol}^{-1}$ ). This energy differential suggests significantly diminished kinetic barriers in the OER process resulting from structural phase modifications. The reduced activation energy conclusively validates the enhanced oxygen evolution reaction efficiency exhibited by the  $\text{IrO}_2/\text{MoS}_2/\text{CNT}$  catalytic system.

**2.4. Mechanism Investigation by Density Functional Theory Calculation.** Based on the experimental findings, correlative optimized theoretical models were constructed, including  $\text{IrO}_2$  (200),  $\text{MoS}_2$  1T (101),  $\text{MoS}_2$  2H (100),  $\text{IrO}_2/\text{MoS}_2$  1T and  $\text{IrO}_2/\text{MoS}_2$  2H (Figures 6a, S21–S27 and the details regarding the selection of the optimal model are provided in the Supporting Information). Interface electronic interactions were analyzed through d-band center theory, where the proximity of metallic d-orbital centers to Fermi levels typically strengthens intermediate adsorption.<sup>40</sup> However, the Sabatier principle dictates that optimal catalytic interfaces require moderated bonding strength, and excessive or insufficient interactions impair reaction dynamics.<sup>41</sup> Projected density of states (PDOS) calculations demonstrated progressive d-band center displacement from  $-0.88 \text{ eV}$  ( $\text{IrO}_2/\text{CNT}$ ) to  $-2.51 \text{ eV}$  ( $\text{IrO}_2/\text{MoS}_2$  2H), then to  $-2.11 \text{ eV}$  ( $\text{IrO}_2/\text{MoS}_2$  1T) (Figure 6b), confirming the electronic modulation effect of  $\text{MoS}_2$  on  $\text{IrO}_2$ . Previous studies associate downward-shifted d-band centers with enhanced  $\text{H}_2\text{O}$  adsorption/dissociation capabilities, improving electrolytic kinetics for water splitting.<sup>42,43</sup> Therefore, the intermediate d-band

position in  $\text{IrO}_2/\text{MoS}_2$  1T suggests superior water-splitting functionality compared to  $\text{MoS}_2$  2H on the catalytic surface.

We further examined the effects of different  $\text{MoS}_2$  phases on the OER catalytic thermodynamics. The geometric configurations of reactive intermediates and associated energy landscapes are detailed in Figures 6c–e and S28. As illustrated in Figure 6d, all four elementary steps (Ir acting as active sites) exhibit uphill free energy changes under the equilibrium conditions ( $U = 0 \text{ V}$ ). Notably, the oxygen evolution bottleneck in  $\text{IrO}_2$  (200) occurs during  $\text{OOH}^*$  to  $\text{O}_2$  conversion, exhibiting a  $2.59 \text{ eV}$  activation barrier - the predominant limitation in acidic OER. Applied potential elevation to  $1.23 \text{ V}$ , the energy barrier of the rate-determining step (RDS) decreases significantly to  $1.36 \text{ eV}$  (Figure 6e). Upon introduction of  $\text{IrO}_2$  into  $\text{MoS}_2$ , the conversion of  $\text{OOH}^*$  to  $\text{O}_2$  for  $\text{IrO}_2/\text{MoS}_2$  1T demonstrates a much lower overpotential of  $1.06 \text{ eV}$  ( $U = 0 \text{ V}$ ), which further decreases to  $0.09 \text{ eV}$  at the operational potential. Phase-dependent mechanistic shifts emerge:  $\text{IrO}_2/\text{MoS}_2$  1T and  $\text{IrO}_2/\text{MoS}_2$  2H systems transition their rate-limiting steps to the formation of  $\text{OOH}^*$ , displaying barriers of  $2.58/2.62 \text{ eV}$  ( $U = 0 \text{ V}$ ) and  $1.35/1.39 \text{ eV}$  ( $U = 1.23 \text{ V}$ ), respectively. The 1T-phase composite achieves superior kinetic enhancement in its third reaction step ( $1.35 \text{ eV}$  barrier), representing the lowest activation energy across tested materials. Crucially, the harmonized energy distribution across reaction coordinates in  $\text{IrO}_2/\text{MoS}_2$  1T promotes both catalytic efficiency and operational stability during prolonged oxygen evolution.

Based on the Sabatier principle, attaining an optimal hydrogen adsorption free energy ( $\Delta G_{\text{H}^*}$ ) close to the

theoretical optimum ( $\Delta G_{H^*} = 0$  eV) is critical for balancing the  $H^*$  adsorption and  $H_2$  desorption.<sup>44,45</sup> Analysis of Figures 6f and S29, the 2H phase of  $MoS_2$  demonstrates the highest  $\Delta G_{H^*}$  value (1.21 eV), reflecting insufficient hydrogen proton adsorption capability at Mo active sites that limits reaction efficiency. The  $IrO_2$ -modified 2H- $MoS_2$  composite shows substantially improved hydrogen adsorption characteristics, with  $\Delta G_{H^*}$  values reaching  $-0.33$  eV at Ir active sites, while the 1T-phase  $IrO_2/MoS_2$  system achieves a marginally more negative  $\Delta G_{H^*}$  value of  $-0.15$  eV. These findings indicate that the 1T-phase  $IrO_2/MoS_2$  heterostructure achieves optimal surface adsorption characteristics for hydrogen intermediates, enabling enhanced HER activity through improved reaction kinetics. The enhanced catalytic performance in both oxygen and hydrogen evolution reactions arises from the synergistic combination of  $IrO_2$  and  $MoS_2$  components, interfacial charge redistribution effects, and superior electrical conductivity within the  $IrO_2/MoS_2/CNT$ s composite system.

**2.5. Overall Water Splitting.** Building upon the exceptional bifunctional catalytic performance of  $IrO_2/MoS_2/CNT$  in oxygen/hydrogen evolution reactions, we developed an acidic water electrolysis system employing symmetrical  $IrO_2/MoS_2/CNT$  electrodes for both the anode and cathode compartments (Figure 7). The water electrolysis performance was evaluated through LSV measurements. As shown in Figure 7b,  $IrO_2/MoS_2/CNT$  demonstrated superior activity, requiring only 1.47 V to achieve  $10\text{ mA cm}^{-2}$  current density, substantially lower than the 1.59 V required by the  $IrO_2||Pt/C$  benchmark system under identical conditions. This performance positions the  $IrO_2/MoS_2/CNT$  catalyst as competitive with, and in some cases superior to, recently reported precious metal-based electrocatalysts in overall water splitting efficiency.<sup>46–48</sup> Given the importance of electrode durability for practical water splitting applications,  $IrO_2/MoS_2/CNT$  was tested for 70 h at  $10\text{ mA cm}^{-2}$  (Figure 7c), during which the cell voltage attenuation is not very noticeable. The nanosheet-like morphology of  $IrO_2/MoS_2/CNT$  was well-preserved after prolonged operation, as shown in Figure 7d, further demonstrating the excellent stability of the as-synthesized material. Additionally, HRTEM (Figure 7e,f) and XPS (Figure 7g,h) analyses confirmed that the surface structure remained largely unchanged after the OER stability tests, underscoring the high durability of  $IrO_2/MoS_2/CNT$  for water splitting.

### 3. CONCLUSIONS

In summary, the  $IrO_2/MoS_2/CNT$  heterostructure featuring controlled phase conversion of  $MoS_2$  from the thermodynamically stable 2H phase to the catalytically active 1T phase alongside an engineered interfacial potential gradient, was successfully fabricated through sequential hydrothermal synthesis. Electrochemical characterization revealed that the optimized catalyst achieves remarkably low HER and OER overpotentials (9 and 182 mV, respectively) at current densities of  $10\text{ mA cm}^{-2}$  in acidic electrolyte conditions. When implemented in a symmetrical electrolyzer configuration, the system required only 1.47 V to reach  $10\text{ mA cm}^{-2}$ , representing a 120 mV improvement over the benchmark  $Pt/C||IrO_2$  configuration. Combined experimental characterization and computational modeling identified iridium centers as principal catalytic sites, with the 1T- $MoS_2$  phase promoting charge redistribution dynamics and interfacial electron transport. This synergistic interaction accelerates hydroxyl inter-

mediate adsorption, optimizes oxygenated species deprotonation pathways, and minimizes activation energy barriers throughout the water dissociation process. The demonstrated phase engineering methodology possesses generalizability potential, establishing a rational design paradigm for developing advanced electrocatalytic systems in renewable energy technologies.

### 4. EXPERIMENTAL SECTION

**4.1. Chemicals.** All reagents used were of analytical-grade purity and were employed as received without additional purification steps. The molybdenum precursor (ammonium molybdate tetrahydrate), sulfur source (thiourea), and iridium precursor ( $IrCl_3 \cdot xH_2O$ ) were sourced from Aladdin Biochemical Technology (Shanghai). Commercial catalysts ( $Pt/C$  and  $IrO_2$ ) were acquired from Johnson Matthey. Nafion solution (5% concentration) and anhydrous ethanol were procured from Sigma-Aldrich. The sulfuric acid electrolyte was supplied by Sinopharm Chemical Reagent Co. All aqueous solutions utilized ultrapure water from Thermo Fisher Scientific (USA).

**4.2. Preparation of  $MoS_2/CNT$  and  $IrO_2/MoS_2/CNT$ .** The  $MoS_2/CNT$  hybrid material was prepared through conventional hydrothermal synthesis following this sequence: Initially, 40 mg of ammonium molybdate and 12 mg of CNTs were dispersed in 10 mL of deionized  $H_2O$  under 30 min of ambient agitation. Subsequently, 0.68 g of thiourea was dissolved in 20 mL of aqueous solution, which was then gradually introduced into the molybdenum-containing suspension during 60 min of continuous mixing. The homogeneous mixture underwent hydrothermal treatment in a Teflon-lined autoclave maintained at  $180\text{ }^\circ\text{C}$  for 24 h. Postreaction purification involved centrifugal separation (8000 rpm, 5 min) followed by sequential ethanol/water rinsing cycles to eliminate ionic residues. Vacuum drying at  $60\text{ }^\circ\text{C}$  for 12 h yielded the intermediate product, which subsequently underwent pyrolysis at  $800\text{ }^\circ\text{C}$  ( $5\text{ }^\circ\text{C/min}$  ramp) under argon flow for 2 h to obtain the final  $MoS_2/CNT$  composite.

The deposition of  $IrO_2$  nanoparticles onto the  $MoS_2/CNT$  substrates was achieved through hydrothermal processing. Precise quantities of the  $MoS_2/CNT$  precursor (15 mg) and iridium precursor (8.56 mg  $IrCl_3 \cdot xH_2O$ ) underwent separate dispersion in 15 mL ethanol aliquots through 60 min of ultrasonication and magnetic agitation. The combined suspension was subjected to hydrothermal treatment in a Teflon-lined reactor at  $100\text{ }^\circ\text{C}$  for 10 h. Postsynthesis processing involved vacuum filtration, solvent rinsing cycles, and final vacuum drying at  $60\text{ }^\circ\text{C}$  for 12 h to obtain the  $IrO_2/MoS_2/CNT$  catalyst.

**4.3. Material Characterization.** Microstructural characterization was conducted utilizing a JEOL-2100F electron microscope operated at 200 kV acceleration voltage, enabling TEM imaging, HRTEM analysis, and elemental distribution mapping via STEM. Crystalline phase identification was performed using a Rigaku D/MAX-2400 X-ray diffractometer with  $Cu\ K\alpha$  irradiation ( $\lambda = 1.5406\text{ \AA}$ ) under 40 kV/100 mA operational parameters. Surface chemical states were examined through XPS measurements performed in ultrahigh vacuum conditions with monochromatic  $Al\ K\alpha$  X-ray excitation (15 kV, 10 mA) and electron energy analysis via a Scienta SES 2002 hemispherical analyzer. Structural fingerprinting was achieved through Raman spectroscopy using a Renishaw in Via system equipped with a 532 nm wavelength excitation source.

**4.4. Electrochemical Measurements.** Electrochemical characterizations employed a conventional three-electrode configuration using a CHI760E potentiostat (CH Instruments). Catalyst inks were prepared by dispersing 4 mg of active material through ultrasonic homogenization in 1 mL of distilled water, followed by precise deposition ( $10\text{ }\mu\text{L}$  aliquot) onto a carbon paper substrate. Electrode preparation was completed by applying 5  $\mu\text{L}$  of Nafion binder (0.2% ethanolic solution) after ink drying. All measurements utilized 0.5 M sulfuric acid electrolyte, with recorded potentials converted to the RHE scale using  $E_{RHE} = E_{SCE} + 0.264\text{ V}$ . Catalytic performance evaluation involved LSV ( $10\text{ mV/s}$  scan rate) for HER/OER characterization under acidic conditions. EIS was acquired across a

0.01–100 kHz frequency spectrum at equilibrium potential. Durability testing comprised 100-h chronopotentiometry at 10 mA cm<sup>-2</sup> and multistep current density profiling (10–50 mA cm<sup>-2</sup> increments) for evaluating electrochemical robustness. Full water electrolysis was implemented through a symmetrical cell configuration where IrO<sub>2</sub>/MoS<sub>2</sub>/CNT functioned as both the anode and cathode. Current–voltage profiles were acquired via LSV (1 mV s<sup>-1</sup> scan rate) in 0.5 M sulfuric acid electrolyte across a 1.2–1.7 V potential window. All data collection occurred without impedance compensation to preserve the inherent system resistance characteristics. Catalytic overpotentials were calculated through the Tafel formulation following  $\eta = a + b \log|j|$ , where  $\eta$  (V) quantifies electrode polarization,  $j$  (mA cm<sup>-2</sup>) reflects operational current density, and  $b$  (mV dec<sup>-1</sup>) characterizes reaction kinetics through Tafel slope analysis. ECSA was determined through double-layer capacitance ( $C_{dl}$ ) measurements using CV across scan rates (10–60 mV s<sup>-1</sup>) within a non-Faradaic voltage range.  $C_{dl}$  values were calculated from the linear regression of capacitive current versus scan rate plots. Turnover frequency (TOF) serves as a fundamental performance indicator for quantifying inherent catalytic efficiency, decoupled from surface area effects. The calculation employs:  $TOF = \frac{j|A|}{m_{Fn}}$ , in which  $j$  represents the current density measured at 300 mV overpotential during LSV testing,  $A$  indicates the geometric electrode area (0.071 cm<sup>2</sup>),  $F$  denotes Faraday's constant (96,485 C mol<sup>-1</sup>), and  $n$  corresponds to active site quantity (mol). The 1/2 factor arises from the two-electron hydrogen evolution pathway, while 1/4 accounts for the four-electron oxygen evolution mechanism.

**4.5. Mott–Schottky (MS) Test.** Semiconductor characterization employed a three-electrode configuration in 0.2 M Na<sub>2</sub>SO<sub>4</sub> aqueous electrolyte, with catalyst-modified indium tin oxide (ITO) used as the working electrode. Catalyst inks were formulated through ultrasonic homogenization (30 min) of 2 mg active material with 100  $\mu$ L isopropanol and 10  $\mu$ L Nafion binder. A 40  $\mu$ L aliquot was deposited onto the ITO substrates (1 cm<sup>2</sup> active area). Vacuum annealing at 60 °C for 12 h achieved electrode stabilization, followed by desiccation storage. Potential values were converted to the RHE scale through  $E_{RHE} = E_{Ag/AgCl} + 0.233$ . Flat-band potentials and carrier densities were derived from capacitance–potential profiles recorded at a 1 kHz frequency.

**4.6. DFT Calculations.** First-principles calculations were conducted utilizing the VASP code within a periodic DFT framework.<sup>49</sup> Electron exchange–correlation effects were modeled through the projector augmented wave–projector augmented wave (GGA-PAW) formalism.<sup>50,51</sup> A 15 Å vacuum spacer was incorporated perpendicular to the catalytic surfaces to eliminate periodic interference. Geometry optimizations employed a 400 eV plane-wave cutoff and  $2 \times 2 \times 1$  k-point sampling using the Monkhorst–Pack grid methodology.<sup>52</sup> Convergence thresholds were set at  $1 \times 1 \times 10^{-5}$  eV for electronic self-consistency and 0.02 eV/Å for ionic force relaxation. Long-range van der Waals interactions were modeled through Grimme's DFT-D3 dispersion correction scheme. Within the computational hydrogen electrode framework, proton–electron pair free energy equivalence to gaseous H<sub>2</sub> was applied.<sup>53</sup> Reaction step energetics were computed as follows:

$$\Delta_G = \Delta_E + \Delta E_{ZPE} - T\Delta S$$

where  $\Delta_E$  denotes energy differences obtained directly from the DFT simulation,  $\Delta E_{ZPE}$  represents zero-point vibrational corrections, and entropy contributions ( $T = 298.15$  K) were derived from standard thermodynamic tables.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnano.4c18288>.

SEM and TEM images of MoS<sub>2</sub>/CNT, TEM images of IrO<sub>2</sub>/CNT, XPS survey for the different samples, UV/vis diffuse reflections, band gap energy results of MoS<sub>2</sub>/

CNT and IrO<sub>2</sub>/CNT, Mott–Schottky plots, XPS valence band spectra, LSV curves of different catalysts for HER, contact angles of the catalyst surfaces of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub>/CNT and MoS<sub>2</sub>/CNT, CV curves for different electrodes, multistep chronopotentiometric curve for the IrO<sub>2</sub>/MoS<sub>2</sub>/CNT electrode, analysis of IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub> by the Laviron equation, LSV curves for IrO<sub>2</sub>/MoS<sub>2</sub>/CNT and IrO<sub>2</sub> and IrO<sub>2</sub>/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub> under both methanol-containing and methanol-free environments, Nyquist plots, Bode phase plots, Arrhenius plots, computational models, OER process of adsorption at each step, and HER process of the model at the \*H adsorption step (PDF)

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## Author Contributions

This work represents a collaborative effort with all coauthors participating in manuscript preparation. The final version received unanimous endorsement from all contributors.

## Notes

The authors declare no competing financial interest.

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